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Solid-state reaction of a CaO–V₂O₅ mixture: A fundamental study for the vanadium extraction process

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Abstract: The aim of this study was to investigate the phase transformation and kinetics of the solid-state reaction of CaO–V₂O₅, which is the predominant binary mixture involved in the vanadium recovery process. Thermal analysis, X-ray diffraction spectroscopy, scanning electron microscopy, and energy dispersive spectrometry were used to characterize the solid-state reaction of the samples. The extent of the solid reaction was derived using the preliminary quantitative phase analysis of the X-ray patterns. The results indicate that the solid reaction of the CaO–V₂O₅ mixture is strongly influenced by the reaction temperature and CaO/V₂O₅ mole ratio. The transformation of calcium vanadate involves a step-by-step reaction of CaO–V₂O₅, CaO–CaV₂O₆, and CaO–Ca₂V₂O₇ depending on the CaO/V₂O₅ mole ratio. The kinetic data of the solid reaction of the CaO–V₂O₅ (1:1) mixture followed a second-order reaction model. The activation energy (E_a) and preexponential factor (A) were determined to be 145.38 kJ/mol, and $3.67 \times 10^8 \text{ min}^{-1}$, respectively.

Keywords: calcium vanadate; vanadium extraction; solid-state reaction; kinetics

1. Introduction

The calcification roasting–acid leaching process, as a clean production method for the extraction of vanadium from metallurgical slags, was first proposed by the Russia Tula factory in the 1970s [1]. This process involves the oxidation roasting of vanadium slag with lime or limestone at approximately 900°C in a multihearth furnace or rotary kiln to convert vanadium compounds to soluble calcium vanadate, followed by the extraction of vanadium from the roasted slag by leaching in dilute sulfuric acid or sodium hydroxide solution, and then the preliminary vanadium product is obtained from the leaching solution via precipitation, solvent extraction, or the ion exchange method [2–3]. Calcification roasting–acid leaching is an environmentally friendly process because it eliminates the use of corrosive gas, has a low additive (lime or limestone) cost, and generates less harmful wastewater and a small amount of noxious solid waste [4–5].

CaO–V₂O₅ is the predominant system for the roasting of vanadium slag with lime or limestone. The phase diagram and thermodynamic data of CaO–V₂O₅ have been reviewed in the literature [6]. The CaO–V₂O₅ system contains four intermediate compounds: metavanadate (CaV₂O₆), pyrovanadate

(Ca₂V₂O₇), orthovanadate (Ca₃V₂O₈), and tetracalcium vanadate (Ca₄V₂O₉) [7]. Among these compounds, CaV₂O₆, Ca₂V₂O₇, and Ca₃V₂O₈ are vital for the vanadium extraction process [8–9]. The solubilities of CaV₂O₆, Ca₂V₂O₇, and Ca₃V₂O₈ in water at 25°C are 0.022, 0.0035, and 0.012 mol/L, respectively [10]; however, they have good solubilities in acidic or alkali solutions. Thus, the transformation of calcium vanadate is of vital importance for the extraction process [5].

Most vanadium is involved in phase with corundum (Fe_{1-x}V_x)₂O₃, the pyroxene M₂Si₂O₆ structure, or spinel M²⁺M₂³⁺O₄ (M²⁺ = Ni, Ca, Mg, Mn, or Fe; M³⁺ = Fe, V, Ti, Cr, or Al) [11]. The transformation of calcium vanadate is strongly influenced by roasting conditions such as the roasting temperature, holding time, atmosphere, additives, heating rate, and so on [12–14]. During roasting, normal vanadium spinel first converts to inverse vanadium spinel at a lower roasting temperature [15]. Then, vanadium spinel and its decomposition product, vanadium oxide, combine with CaO to generate CaV₂O₆ at low temperature and Ca₂V₂O₇ at approximately 630°C. Ca₃V₂O₈ can be generated with an excess amount of CaO above 750°C [10]. At a low CaO content in the slag, the system undergoes a CaO deficiency for

complete fixation of V₂O₅ into calcium vanadate. At a high CaO content, vanadium partly dissolves in perovskite, which does not undergo structural changes upon roasting [16]. A lower heating rate was found to promote the conversion of vanadium and make the optimum roasting temperature shift to the high temperature region [1]. Furthermore, the calcification roasting process can be enhanced using microwave heating [17] and mechanical activation pretreatment [18] and by improving the oxygen partial pressure [9].

The formation of calcium vanadate is thermodynamically favorable [19]; however, the actual process is usually limited by the dynamic conditions [1]. Therefore, the kinetic data for the solid–solid reaction of a CaO–V₂O₅ mixture are very important for industrial applications. The present investigation was therefore undertaken to clarify the phase evolution of the CaO–V₂O₅ mixture at different temperatures. The conversion degree of the solid-state reaction of the CaO–V₂O₅ mixture was determined using X-ray quantitative phase analysis and the solid–solid reaction kinetics were examined.

2. Experimental

2.1. Materials

The CaO and V₂O₅ used in the experiments were of analytical grade with a purity greater than 99.9% and were preheated at 900 and 500°C, respectively, for 5 h to remove free water and crystal water. The samples were cooled to approximately 100°C in a furnace and further cooled to room temperature in a dryer with allochroic silica.

2.2. Experimental procedure

A determined amount of CaO and V₂O₅ with varied mole ratios was thoroughly ground and mixed in an agate mortar. The thermal analysis of the mixture was conducted on a simultaneous thermogravimetry and differential scanning calorimetry (TG–DSC; STA 449C, NETZSCH, Germany). Approximately 15 mg of the mixture was loaded in an alumina crucible and heated from 30 to 800°C at a constant rate of 2°C/min. Air (21vol% O₂ and 79vol% N₂) was flowed into the furnace at a rate of 50 mL/min during heating. Each experiment was performed twice for reproducibility and exact thermograms were obtained. The solid reaction experiments were performed in a muffle furnace and regulated to $\pm 2^\circ\text{C}$ via a temperature programmed control instrument. A determined amount of CaO–V₂O₅ mixture was roasted at a determined temperature for different times by placing the cold alumina crucible into a preheated muffle furnace and removing it while hot. In this way, the heating and cooling rates were kept approximately equal.

2.3. Analytical methods

The mineralogical composition of the samples was identified via X-ray powder diffraction (XRD) using an X-ray dif-

fractometer (PANalytical X'Pert Powder, PANalytical B. V.) with Cu K α radiation ($\lambda = 0.154$ nm, 40 kV, 75 mA) at room temperature. The 2θ scan range was from 10° to 60° at a rate of 0.01°/s. The phase analysis of the X-ray diffractograms was performed using the search/match software program MDI Jade 6.0 applied to the ICDD Powder Diffraction File database. The preliminary quantitative phase analyses were assessed using the Rietveld method [20]. This method creates an approximate model of the diffractogram based on input parameters. This model is then refined by the least squares method in iterative steps until the best possible approximation to the measured pattern is found. The scanning electron microscopy (SEM)/energy dispersive X-ray spectroscopy (EDS) analysis of the samples was performed using a scanning electron microscope (VEGA 3 LMH; TESCAN, Czech Republic) equipped with an energy dispersive X-ray spectroscope (Oxford, UK).

3. Results and discussion

3.1. TG–DSC analysis

The TG and DSC curves for the CaO–V₂O₅ (1:1, mole ratio, so as the follows) mixture are provided in Fig. 1, which shows that the weight loss up to 402°C is 4.47%, corresponding to the release of adsorbed water molecules in the mixture. The release of these water molecules produces the corresponding endothermic peak in the DSC curves over the same temperature ranges. A weight loss of 0.87% is observed in the range of 402–600°C, which should correspond to the volatilization of V₂O₅; however, the weight of the sample is almost constant at above 600°C. Two sharp endothermic peaks are detected at 623.4 and 769.7°C that can be attributed to the fusion of V₂O₅ and CaV₂O₆, respectively.

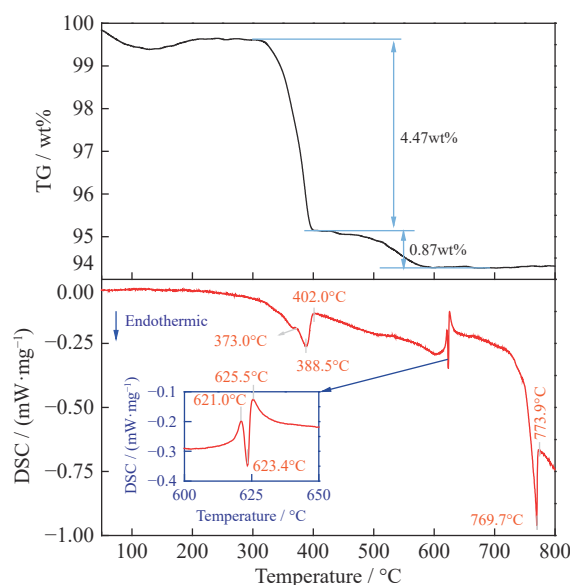


Fig. 1. TG–DSC analysis of the CaO–V₂O₅ (1:1) mixture.

These fusion data are consistent with the reported peaks at 630 and 770°C [21].

3.2. Phase transformation

Fig. 2 shows the XRD patterns of the products of the CaO–V₂O₅ (1:1) mixture roasted at 400, 450, 500, 550, and 600°C for 40 h. As seen in Fig. 2, three major phases are apparent from the XRD patterns, namely, CaO, V₂O₅, and CaV₂O₆. The intensity of V₂O₅ and CaO gradually decreased with the increase in temperature from 400 to 500°C and continued to decrease slightly from 500 to 600°C. In the meantime, the intensity of CaV₂O₆ gradually increased with an increase in temperature. No other intermediate compounds were found under these experimental conditions.

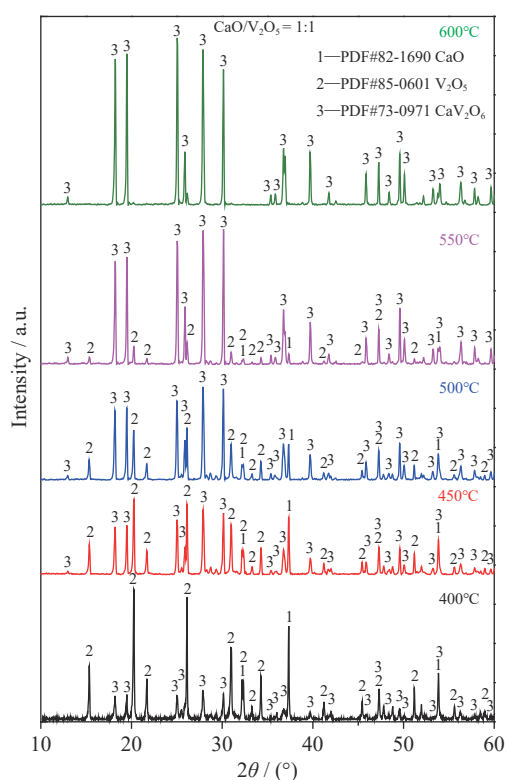


Fig. 2. XRD patterns of the products of the CaO–V₂O₅ (1:1) mixture roasted at different temperatures.

Fig. 3 shows the preliminary quantitative analysis results of the X-ray patterns of the CaO–V₂O₅ mixtures with different CaO/V₂O₅ mole ratios roasted at different temperatures. As shown in Fig. 3(a), for the CaO–V₂O₅ mixture with a CaO/V₂O₅ mole ratio of 1:1, the content of CaV₂O₆ steadily increased from 22% to approximately 99% as the temperature increased from 400 to 600°C.

The XRD patterns of the products of the CaO–V₂O₅ (2:1) mixture roasted at 400, 450, 500, 550, 600, and 650°C for 40 h revealed that the transformation of the CaO–V₂O₅ mixture with a mole ratio of 2:1 involves the formation of CaV₂O₆ and Ca₂V₂O₇. The XRD patterns of the roasted samples are

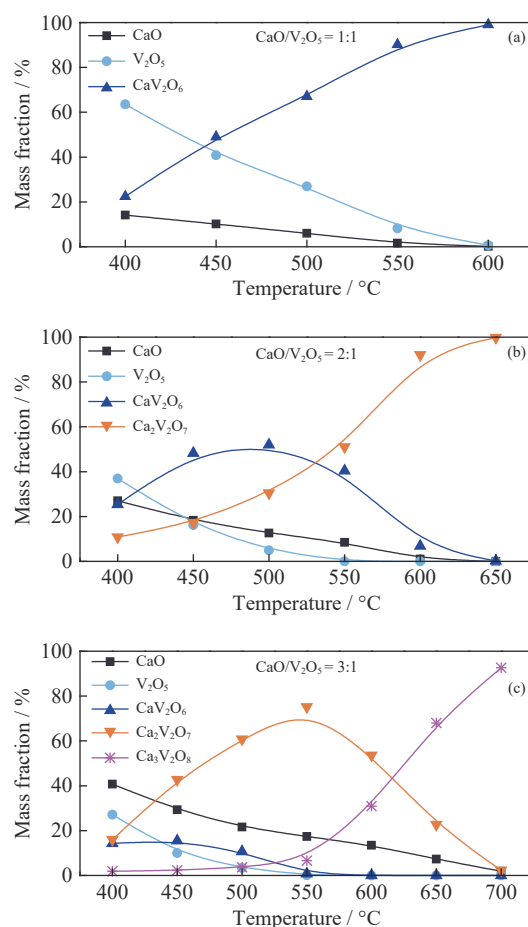


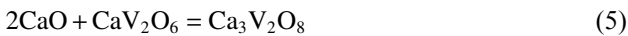
Fig. 3. Variation of the phase compositions of the CaO–V₂O₅ mixtures with different CaO/V₂O₅ mole ratios as a function of roasting temperature: (a) 1:1; (b) 2:1; (c) 3:1.

not shown here. As shown in Fig. 3(b), for the CaO–V₂O₅ mixture with a mole ratio of 2:1, the content of CaO and V₂O₅ substantially decreased with the increase in temperature and reached to zero at 650 and 550°C, respectively. However, the content of CaV₂O₆ initially increased rapidly, reached a maximum value at 500°C, and then decreased sharply at higher temperature (500–600°C), and eventually reduced close to zero at 650°C. The content of Ca₂V₂O₇ slightly increased at lower temperature (400–500°C), and then substantially increased at higher temperature (550–650°C) because of the solid-state reaction of CaV₂O₆ and CaO.

For the CaO–V₂O₅ mixture with a mole ratio of 3:1, as shown in Fig. 3(c), the main phases of the sample roasted at 400°C are CaO, V₂O₅, CaV₂O₆, and Ca₂V₂O₇ with a small amount of Ca₃V₂O₈. When the temperature increased from 400 to 550°C, the contents of CaO and V₂O₅ significantly decreased, whereas the content of CaV₂O₆ slightly increased initially and then decreased and almost disappeared at 550°C; however, the content of Ca₂V₂O₇ increased rapidly and reached a maximum value at 550°C. When the temperature increased from 550 to 700°C, the content of Ca₂V₂O₇ and

CaO substantially decreased, while the content of Ca₃V₂O₈ significantly increased. This indicates that the solid reaction of CaV₂O₆ and CaO is basically completed at 550°C with a final product of Ca₂V₂O₇, whereas the solid reaction of Ca₂V₂O₇ and CaO is completed at 700°C with a final product of Ca₃V₂O₈.

The solid-state reactions between CaO and V₂O₅ have been partly described previously [1,16,22–23]. CaV₂O₆, Ca₂V₂O₇, and Ca₃V₂O₈ may be formed by the following reactions:



A plot of the standard Gibbs free energy (ΔG^\ominus) as a function of temperature for the above reactions is shown in Fig. 4. The ΔG^\ominus for all the reactions is negative at low temperatures, which indicates that the above reactions are thermodynamically favorable. The values of ΔG^\ominus are more negative for reaction (1) than for reactions (4) and (6) below 600°C, indicating that the formation of calcium vanadate follows the order CaV₂O₆ > Ca₂V₂O₇ > Ca₃V₂O₈. The variation of the contents of CaV₂O₆, Ca₂V₂O₇, and Ca₃V₂O₈, presented in Fig. 3, confirm this finding.

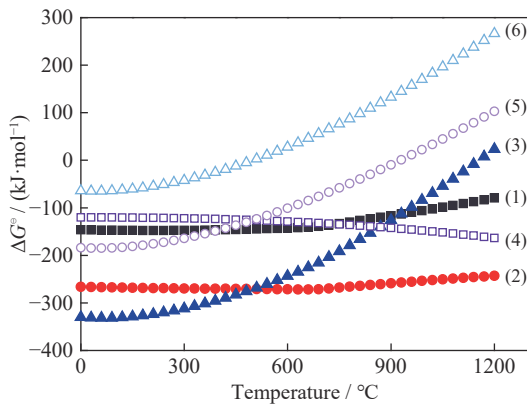


Fig. 4. ΔG^\ominus – T plots for the possible reactions of CaO–V₂O₅ mixture.

3.3. Solid-state reaction kinetics

Examination of the solid-state reaction kinetics of the CaO–V₂O₅ system using TG–DSC is very difficult because there should be no theoretical weight change except the release of water molecules and the volatilization of V₂O₅ during the heating process (as shown in Fig. 1). Furthermore, the enthalpy changes in the solid-state reaction of CaO–V₂O₅ system are small and complex, and therefore difficult to dis-

tinguish from noise and the heat of the phase transformation. Therefore, the solid-state reaction kinetics of the CaO–V₂O₅ system was investigated by examining the phase variation of the samples roasted at different temperatures and holding times. The formation of CaV₂O₆ is the basis of the formation of other types of calcium vanadates and the most important intermediate product in the vanadium extraction process. Therefore, the solid-state reaction kinetics of the representative CaO–V₂O₅ (1:1) mixture was investigated. The XRD patterns of the products of the CaO–V₂O₅ (1:1) mixture roasted at 450, 500, and 550°C for different times are performed (figure not given here). The XRD diffraction intensity of V₂O₅ gradually decreased with the increase of roasting temperature and the extending of reaction time, whereas the XRD diffraction intensity of CaV₂O₆ gradually increased.

The preliminary quantitative phase analysis of the XRD patterns was used to determine the mass fraction of CaV₂O₆. The variation of CaV₂O₆ in the CaO–V₂O₅ mixture with reaction temperature and time is shown in Fig. 5. The formation rate of CaV₂O₆ substantially increased with the increase in temperature from 450 to 500°C and slightly increased with a further increase to 550°C. Furthermore, the yield of CaV₂O₆ increased almost linearly with time at 450°C, while it initially increased exponentially and then increased slightly at 500 and 550°C.

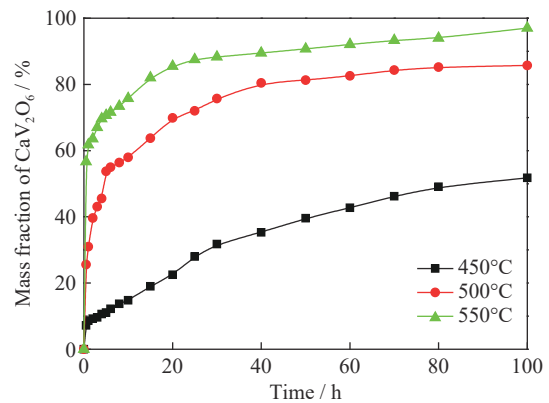


Fig. 5. Mass fraction of CaV₂O₆ in the CaO–V₂O₅ (1:1) mixture roasted at 450, 500, and 550°C for different times.

The heterogeneous solid-state reaction is generally expressed as [24],

$$\frac{d\alpha}{dt} = k(T) f(\alpha) \quad (7)$$

$$\alpha = \frac{w}{100} \quad (8)$$

where α is the degree of conversion, w is the mass percentage of product, t is the time, and T is the temperature. The reaction rate constant $k(T)$ follows the Arrhenius expression:

$$k(T) = A \exp\left(-\frac{E_a}{RT}\right) \quad (9)$$

Combining Eqs. (7) and (9) gives

$$\frac{d\alpha}{dt} = A \exp\left(-\frac{E_a}{RT}\right) f(\alpha) \quad (10)$$

In Eqs. (9) and (10), R is the gas constant and A , E_a , and $f(\alpha)$ are the so-called “kinetic triplet.” A is the preexponential factor, E_a is the activation energy, and $f(\alpha)$ is a differential function that represents the reaction model. Integration of Eq. (10) leads to another integral function that represents the kinetic model:

$$g(\alpha) = \int_0^\alpha \frac{d\alpha}{f(\alpha)} = A \int_0^t \exp\left(-\frac{E_a}{RT}\right) dt \quad (11)$$

The kinetic parameters can be obtained using the model-free approach method or the model fitting method. The model-free approach method in isothermal kinetics helps to avoid the problems that originate from the ambiguous evaluation of the reaction model [25]; however, it needs continuous experimental data points. The experimental data were applied to the most commonly used kinetic reaction models such as reaction order, diffusion, nucleation, and geometrical contraction. According to the model fitting results, the second-order reaction model has the highest regression coefficients at all temperatures and can therefore be considered more suitable for describing the kinetics of the solid-state reaction of the CaO–V₂O₅ mixture. The straight lines shown in Fig. 6 indicate that the second-order model can well represent the solid reaction process. Thus, the equation representing the kinetics of this process can be written as

$$g(\alpha) = (1 - \alpha)^{-1} - 1 = kt \quad (12)$$

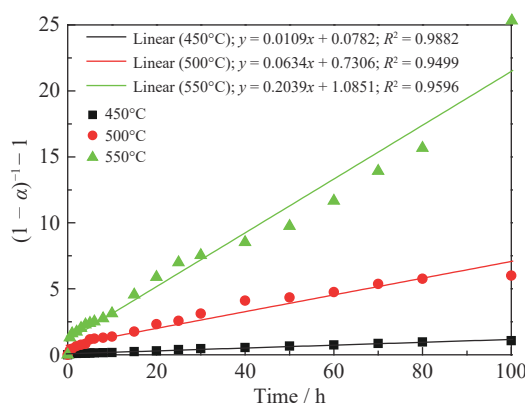


Fig. 6. Variation of $(1 - \alpha)^{-1} - 1$ with time for the solid reaction of the CaO–V₂O₅ (1:1) mixture.

The activation energy of the reaction can be determined using the Arrhenius equation. Taking the natural logarithm on both sides of Eq. (9) [26],

$$\ln k = \ln A - \frac{E_a}{RT} \quad (13)$$

When the reaction obeys the Arrhenius equation, a plot of $\ln k$ versus $1/T$ is a straight line with slope $-E_a/R$ and intercept $\ln A$. The Arrhenius plot of $\ln k$ versus $1/T$, shown in

Fig. 7, gives an apparent activation energy of 145.38 kJ/mol and a preexponential factor A of $3.67 \times 10^8 \text{ min}^{-1}$ for the solid reaction of the CaO–V₂O₅ (1:1) mixture.

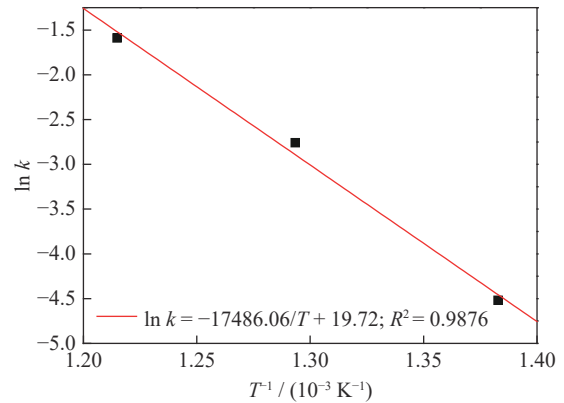


Fig. 7. Arrhenius plot for the solid reaction of the CaO–V₂O₅ (1:1) mixture.

3.4. SEM/EDS analysis

Fig. 8 shows the back-scattered electron (BSE) micrographs of the CaO–V₂O₅ (1:1) mixture roasted at 450°C for 100 h. The image shows two types of particles with different BSE intensities agglomerated together, a bright white one and a gray one. The bright white particles are approximately spherical or ellipsoidal with a diameter of 2–4 μm , whereas the gray particles are amorphous with a wide range of sizes (Figs. 8(a) and 8(b)). Figs. 8(c)–8(f) show the elemental distribution maps of V, Ca, O, and C for the region marked by a white square in Fig. 8(b). Ca is obviously enriched in the gray particles, whereas V is enriched in the white particles. These results indicate that the solid-state reaction of V₂O₅ and CaO is incomplete under the roasting conditions of 450°C and 100 h.

The BSE micrographs of the CaO–V₂O₅ mixture roasted at 550°C for 100 h are shown in Fig. 9. No samples with significantly different gray levels are observed in Figs. 9(a) and 9(b). Compared with the samples roasted at 450°C, the morphology of the particles roasted at 550°C tends to be uniform. The particles are almost spherical and obviously sintered together (Fig. 9(b)). As shown in Figs. 9(c)–9(f), the EDS patterns show that points 1 to 4 in Fig. 9(b) all enriched with Ca, V, and O. Furthermore, the atomic proportions of Ca, V, and O are close to that of CaV₂O₆. This result means that the solid reaction of the CaO–V₂O₅ mixture ran to completion under the roasting conditions of 550°C and 100 h. The change in the sample morphology is basically consistent with the study on calcification roasting of vanadium slag. With an increase in temperature (660–900°C), the vanadate transforms in morphology from a small amount of dissemination to microparticles and an amorphous molten zone and migrates to the outside of the slag particles [27].

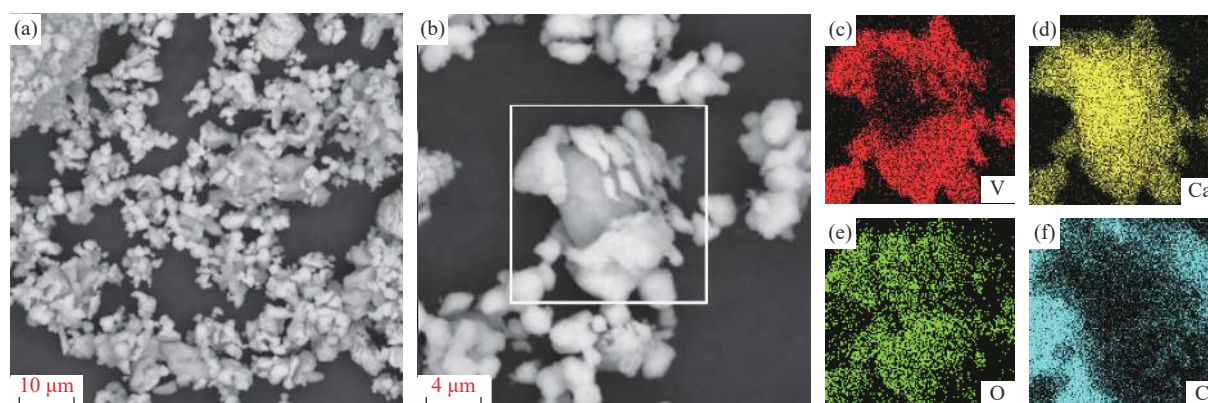


Fig. 8. (a, b) Back-scattered electron (BSE) micrographs of the CaO–V₂O₅ (1:1) mixture roasted at 450°C for 100 h; elemental distribution maps of the region marked by a white square in (b): (c) V; (d) Ca; (e) O; (f) C.

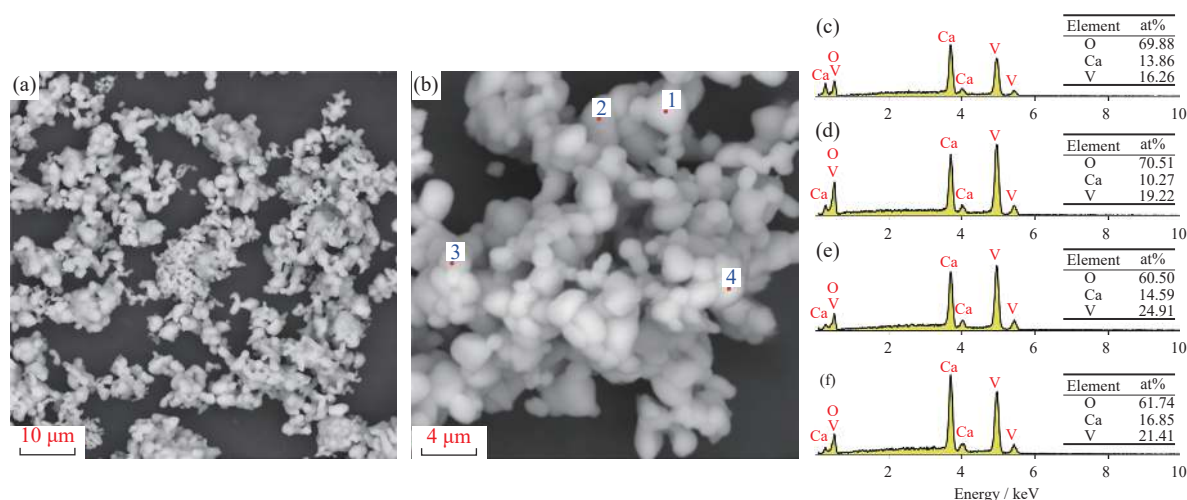


Fig. 9. (a, b) Back-scattered electron (BSE) micrographs of the CaO–V₂O₅ (1:1) mixture roasted at 550°C for 100 h; EDS patterns of points 1–4 in (b): (c) point 1; (d) point 2; (e) point 3; (f) point 4.

4. Conclusions

The phase transformation and solid-state reaction kinetics of a CaO–V₂O₅ mixture were investigated in the present study. The following conclusions can be drawn.

(1) The solid reaction of the CaO–V₂O₅ mixture is strongly influenced by the reaction temperature and CaO/V₂O₅ mole ratio. The transformation of calcium vanadates goes through a step-by-step solid-state reaction of CaO–V₂O₅. The formation of CaV₂O₆ involves the simple solid reaction of CaO–V₂O₅. The generation of Ca₂V₂O₇ involves the formation of CaV₂O₆ and the reaction of CaV₂O₆–CaO. However, the formation of Ca₃V₂O₈ involves the formation of CaV₂O₆ and CaV₂O₇. Only the CaV₂O₆, Ca₂V₂O₇, and Ca₃V₂O₈ phases can be detected at 600, 650, and 700°C for the solid-state reaction of the CaO–V₂O₅ mixture with a CaO/V₂O₅ mole ratio of 1:1, 2:1, and 3:1, respectively.

(2) The thermal analysis shows that the fusion of V₂O₅ and CaV₂O₆ occurred at 623.4 and 769.7°C, respectively. The solid reaction of the CaO–V₂O₅ (1:1) mixture is strongly in-

fluenced by the reaction temperature and time. The mass fraction of CaV₂O₆ linearly increased with holding time at 450°C, whereas it substantially increased initially and then remained roughly stable above 500°C. With the increase in temperature from 450 to 550°C at a reaction time of 100 h, the samples changed from parts of unreacted material with irregular shapes to the CaV₂O₆ product with an approximately uniform and spherical morphology.

(3) Graphical and statistical methods show that the reaction of CaO–V₂O₅ (1:1) can be described using a second-order reaction model in the form of $(1 - \alpha)^{-1} - 1 = kt$. The E_a and A for the solid reaction of CaO–V₂O₅ (1:1) were determined to be 145.38 kJ/mol and $3.67 \times 10^8 \text{ min}^{-1}$, respectively.

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