

## Fabrication and characterization of anodic oxide films on a Ti-10V-2Fe-3Al titanium alloy

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**Abstract:** Anodic oxide films of the titanium alloy Ti-10V-2Fe-3Al in ammonium tartrate electrolyte without hydrofluoric acid or fluoride were fabricated. The morphology, components, and microstructure of the films were characterized by scanning electron microscopy (SEM), X-ray photoelectron spectroscopy (XPS), X-ray diffraction (XRD), and Raman spectroscopy. The results showed that the films were thick, uniform, and nontransparent. Such films exhibited sedimentary morphology, with a thickness of about 3  $\mu\text{m}$ , and the pore diameters of the deposits ranged from several hundred nanometers to 1.5  $\mu\text{m}$ . The films were mainly titanium dioxide. Some coke-like deposits, which may contain or be changed by OH, NH, C–C, C–O, and C=O groups, were doped in the films. The films were mainly amorphous with a small amount of anatase and rutile phase.

**Key words:** titanium alloys; anodic oxide films; titanium dioxide; pulse current method

### 1. Introduction

Titanium is a corrosion-resistant metal with applications in several fields, such as aerospace, chemical, and electrochemical industries [1-2]. It is also used for biomedical and dental implants [3-5]. The favorable corrosion resistance of titanium should be attributed to the anodic oxide film naturally formed on its surface [6]. Corrosion is believed to occur through “weak spots” in the natural oxide film; a forced increase in film thickness will eliminate such weak spots and increase the corrosion resistance. Anodic oxidation is a commonly used surface treatment on the titanium substrate to increase the thickness and the corrosion resistance of the film [7]. The properties of anodic oxide films strongly depend on their composition, structure, and thickness [8]. Therefore, several articles have been devoted to the studies of such characteristics of the films in different preparation conditions [9-11].

Ti-10V-2Fe-3Al is an excellent candidate for aerospace applications [1, 12] owing to its excellent properties. However, there are very few reports on the anodic oxide films of Ti-10V-2Fe-3Al [13].

Generally, the films are almost several tens of nanometers [14] to about 1  $\mu\text{m}$  [15] in size. Such films are transparent and the colors of the films have been

explained by the multiple-beam interference theory [16]. Jeong *et al.* [17] used hydrofluoric acid as the pretreatment reagent and prepared a 5-7- $\mu\text{m}$  thick film at high voltage. Schmuki *et al.* [18-19] fabricated a thick anodic oxide film with the thickness of 2.5-7  $\mu\text{m}$ , however, the electrolyte they used contained hydrofluoric acid [18] or fluoride [19]. Hydrofluoric acid and fluoride are harmful to the environment owing to their high toxicity and complexity of their disposal process.

A novel anodic oxidizing process on Ti-6Al-4V titanium alloys with Ce element was studied in the previous article [20]. In this article, the thick, uniform, and nontransparent anodic oxide films of titanium alloy Ti-10V-2Fe-3Al were fabricated in the ammonium tartrate electrolyte without hydrofluoric acid or fluoride. The morphology, components, and microstructure of the anodic oxide films were investigated by scanning electron microscopy (SEM), X-ray photoelectron spectroscopy (XPS), X-ray diffraction (XRD), and Raman spectroscopy.

### 2. Experimental

#### 2.1. Fabrication of anodic oxide films

A Ti-10V-2Fe-3Al forged block was cut into the

sheets of 80 mm×25 mm×2 mm. The nominal composition (wt%) of Ti-10V-2Fe-3Al is V 10.100, Fe 20.100, Al 3.100, C <0.050, O <0.130, and Ti balanced. Each sample was abraded with the silicon carbide (SiC) paper of successive grades from 300 to 600 grit, followed by rinsing with acetone and deionized water successively and finally dried in air.

Anodic oxidation was carried out in a cell with a thermostat water bath and a magnetic stirring apparatus. The sample was used as the anode and a 1Cr18Ni9Ti stainless steel plate was used as the cathode in an electrolytic cell. The anode surface was less than 50% of that of the cathode. The electrolyte used was an aqueous solution of ammonium tartrate, which was prepared from analytical grade chemicals and deionized water. The sample suspended by a copper wire was partly immersed into the electrolyte, which ensured that the copper wire did not contact with the electrolyte. The area under the water was calculated before anodic oxidation for setting the current density.

Anodic oxidation was performed using a pulse galvanostatic power supply WMY-IV. The fabrication conditions were studied in detail previously [21]. The main parameters are shown in Table 1.

**Table 1. Parameters of the fabrication process**

Content of ammonium tartrate	3 g/L
Current density	10 A/dm <sup>2</sup>
Duty ratio	30%
Frequency	100/min
Temperature	15±2°C
Agitation speed	100 r/min

## 2.2. Characterization of anodic oxide films

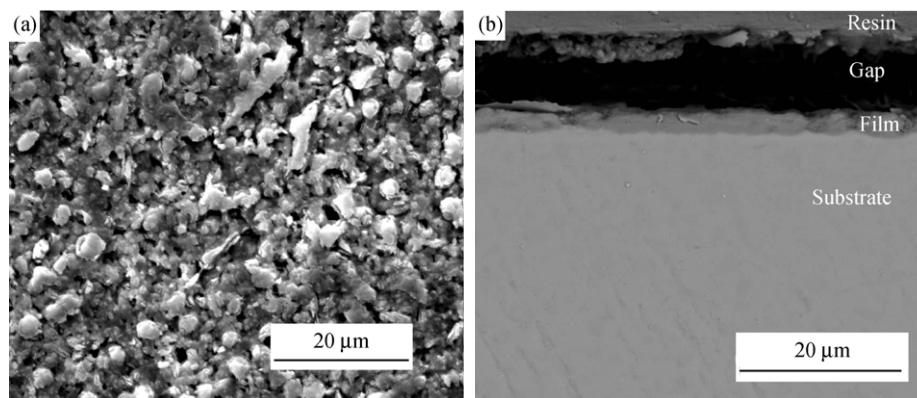
The sample morphology was observed by SEM (FEI-Quanta600D8032). The components of the sample surface were analyzed by XPS (Kratos-Axis Ultra, using Al K<sub>α</sub> radiation at 225 W, 15 mA, 15 kV, C1s hydrocarbon peak at 284.80 eV). The microstructure of the anodic oxide films was determined by XRD (Rigaku-D/Max2200PC, using Cu K<sub>α</sub> radiation at 40 kV, 40 mA, and at the scanning speed of 4°/min). Additional information was also obtained by Raman spectroscopy (Yvon Jobin Horiba-HR800, using a He-Ne laser without filter, 633 nm).

## 3. Results and discussion

### 3.1. Morphology of anodic oxide films

The films fabricated were uniform and nontransparent yellow. The sedimentary morphology of the anodic oxide films formed can be seen clearly in Fig. 1(a). The pores have diameters ranging from several hundred nanometers to 1.5 μm, and there are some particles on the surface. It is likely owing to the adsorption or deposition of some substances from the electrolyte, or to the fragments of the films. The ratification is demonstrated by the subsequent XPS results.

Fig. 1(b) shows a cross-sectional image of the anodic oxide film. The thickness of such a film is around 3 μm, which indicates that it is relatively thick, compared to the anodic oxide films fabricated on titanium and its alloys by other processes without hydrofluoric acid or fluoride.



**Fig. 1. Scanning electron images of the surface (a) and the cross-section (b) of anodic oxide films.**

### 3.2. Components of anodic oxide films

The survey spectrum of anodic oxide films is presented in Fig. 2. It can be observed that the anodic oxide films mainly contain Ti, O, C, and N elements. The photoelectron peaks for Ti2p, O1s, C1s, and N1s appear clearly at the binding energies of 458.25, 530.01, 284.80, and 399.95 eV, respectively. The peak

positions are in agreement with the reference values, which provide the O1s and Ti2p binding energies relative to the TiO<sub>2</sub> species of 530 and 458.6 eV, respectively [22].

Fig. 3 shows the high resolution XPS spectrum of the Ti2p region of anodic oxide films. The Ti2p photoemission signal splits in two peaks, the first at

the energy of 458.35 eV, attributed to  $\text{Ti}2p_{3/2}$  (IV) at  $458.8 \pm 0.1$  eV, and the second at the energy of 464.04 eV, attributed to  $\text{Ti}2p_{1/2}$  (IV) at  $464.5 \pm 0.1$  eV [8], showing that  $\text{TiO}_2$  is the main constituent of the anodic oxide films.

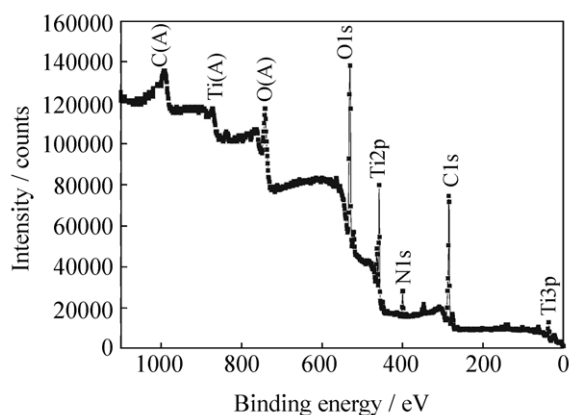


Fig. 2. XPS survey spectrum of anodic oxide films.

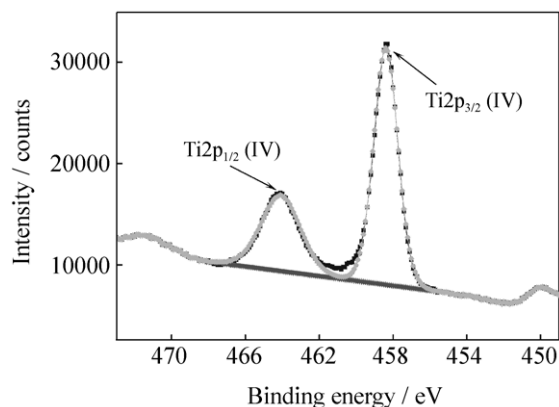


Fig. 3. XPS spectrum of the  $\text{Ti}2p$  region of anodic oxide films.

Fig. 4 shows the high resolution XPS spectrum of the  $\text{O}1s$  region of anodic oxide films. The  $\text{O}1s$  photoemission signal splits in three peaks, the first at the energy of 529.77 eV, attributed to  $\text{Ti-O}$  bonds at  $(530.5 \pm 0.2)$  eV, the second at the energy of 531.42 eV, attributed to  $\text{OH}$  bonds at  $(531.8 \pm 0.1)$  eV, and the third at the energy of 532.72 eV, attributed to adsorbed  $\text{H}_2\text{O}$  at  $(532.5 \pm 0.2)$  eV [8].

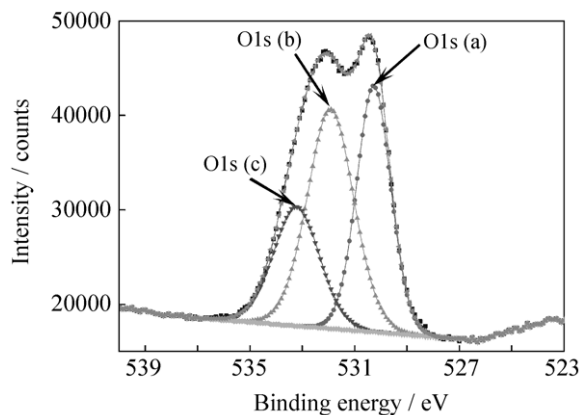


Fig. 4. XPS spectrum of the  $\text{O}1s$  region of anodic oxide films.

The high resolution XPS spectrum of the  $\text{N}1s$  region of anodic oxide films is shown in Fig. 5. The only peak is at 399.99 eV, which is attributed to the  $\text{NH}$  group [23], corresponding to the electrolyte.

Fig. 6 shows the high resolution XPS spectrum of the  $\text{C}1s$  region of anodic oxide films. Three peaks occur at 284.80, 286.28, and 288.07 eV for the prepared films. The first value arises from the adventitious elemental carbon ( $\text{C-C}$ ), and the other two small peaks indicate the existence of  $\text{C-O}$  and  $\text{C=O}$ . This finding agrees well with the previous studies [24]. These groups are corresponding to the electrolyte. The molecular structural formula of ammonium tartrate is as follows.

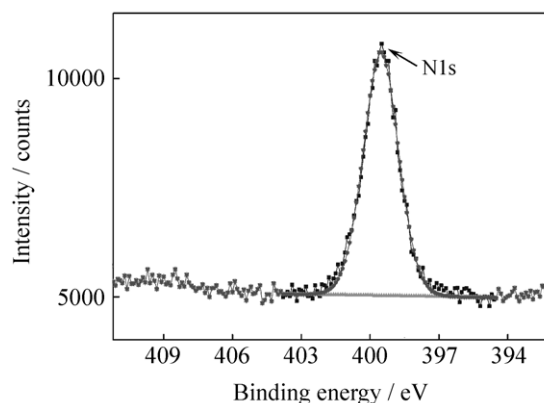
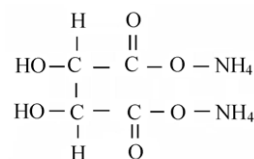


Fig. 5. XPS spectrum of the  $\text{N}1s$  region of anodic oxide films.

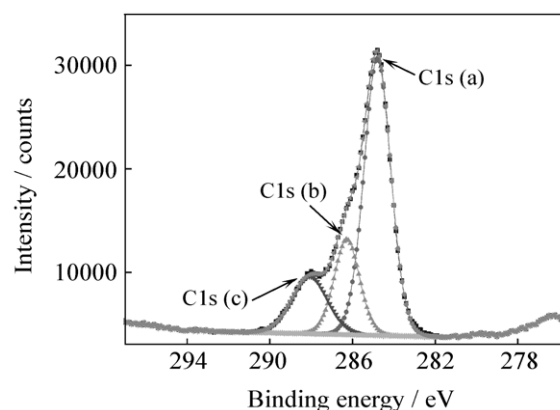


Fig. 6. XPS spectrum of the  $\text{C}1s$  region of anodic oxide films.

An ammonium tartrate molecule contains  $\text{OH}$ ,  $\text{NH}$ ,  $\text{C-C}$ ,  $\text{C-O}$ , and  $\text{C=O}$  groups, which indicates that ammonium tartrate may absorb in the pores of the films and deposit after water volatilization, or be involved in the film formation reactions. The amount of carbon on the  $\text{TiO}_2$  surface is responsible for good

visible-light absorbance. This may be attributed to the phenomenon that the film is nontransparent. A few oxygen atoms on the  $\text{TiO}_2$  surface may be substituted by carbon to form a  $\text{TiO}_{2-x}\text{C}_x$  structure. The XPS analysis of  $\text{Ti}2p_{3/2}$  does not immediately reveal the Ti–C bond, and therefore, the  $\text{TiO}_{2-x}\text{C}_x$  structure of these samples is probably scarce. Several OH groups transformed to highly condensed species, like coke, and were deposited and doped into anodic oxide films [25]. This is consistent with the ratiocination for the sedimentary morphology of anodic oxide films. Some sites are covered by coke species, causing an incomplete lattice structure (amorphous-like surface structure) on the anodic oxide films. Therefore, carbon in the films may play two roles: it is the sensitizer for visible-light absorption owing to its coke-like structure; it also exists as an impurity, which makes the lattice defect of  $\text{TiO}_2$  form interface states that effectively lower the band gap.

### 3.3. Microstructure of anodic oxide films

Fig. 7 shows the XRD spectra of the anodic oxide films formed on Ti-10V-2Fe-3Al (a) and the oxide powder stripped from the Ti-10V-2Fe-3Al substrate (b). The peaks shown in Fig. 7(a) are related to the substrate. The typical small broad peak  $\alpha$  in Fig. 7(b) indicates the presence of the amorphous structure [26]. The small broad peak  $\alpha$  corresponding to the (101) peak  $25.4^\circ$  of anatase is observed, whereas the peak  $\beta$  refers to the (103) and (004) peaks of anatase. Peak  $\chi$  is related to the (200) peak of anatase, and peak  $\delta$  is associated with the (105) and (211) peaks of anatase [17]. The microstructure of the anodic oxide films in this article is different compared to that of the previous references owing to different compositions and thermal histories of the metallic substrate, electrolytes, and fabrication parameters. The crystal phase is not clearly seen in the spectra of the anodic films. Since (i) a relatively low temperature ( $15 \pm 2^\circ\text{C}$ ) cannot lead to amorphous to anatase and rutile crystal transformation, (ii) the alloying components of the Ti-10V-2Fe-3Al substrate inhibit the crystal nucleation, and (iii) the incorporation of ammonium and tartrate species into anodic oxide films is associated with the amorphous-to-crystalline transition, it is reasonable to conclude that such a low crystal composition in the anodic films is too small to be seen in the XRD spectra.

In order to obtain further information on the crystal structure on the surface of the films, the Raman spectrum of such films is shown in Fig. 8. In this spectrum, the two bands positioned at  $380\text{--}450$  and  $600\text{--}650$   $\text{cm}^{-1}$  are corresponding to the characteristic peaks of anatase phase (E1g mode  $399$   $\text{cm}^{-1}$ , Eg mode  $640$

$\text{cm}^{-1}$ ) and rutile phase (Eg mode  $445$   $\text{cm}^{-1}$ , A1g mode  $607$   $\text{cm}^{-1}$ ). These bands were already observed in the suspension plasma sprayed  $\text{TiO}_2$  coatings [27]. The appearance of the broad bands may be attributed to the effect of impurity doping, defects, and strains introduced from the pretreatment process [28]. The intense peak at  $158$   $\text{cm}^{-1}$  is also corresponding to rutile (B1g mode  $144$   $\text{cm}^{-1}$ ) or anatase (Eg mode  $147$   $\text{cm}^{-1}$ ) [29]. This blue-shift peak of the anodic films is caused by a low degree of crystallization on the surface and small particles [25].

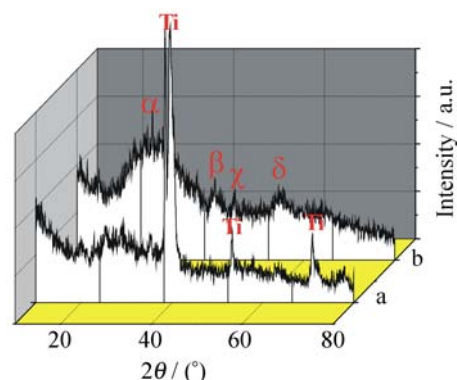


Fig. 7. X-ray spectra of the anodic oxide films (a) and the oxide powder (b).

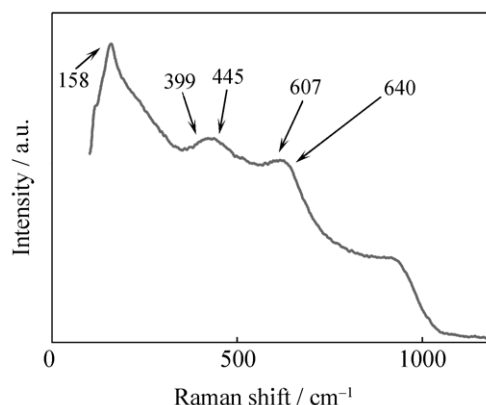


Fig. 8. Raman spectrum of anodic oxide films.

## 4. Conclusion

The thick, uniform, and nontransparent anodic oxide films of the titanium alloy Ti-10V-2Fe-3Al in ammonium tartrate electrolytes without toxic hydrofluoric acid and fluoride were successfully fabricated. The SEM results showed that some substances were adsorbed in or deposited on the films, exhibiting a sedimentary morphology with the film thickness of about  $3$   $\mu\text{m}$  and pore diameters of the deposits ranging from several hundred nanometers to  $1.5$   $\mu\text{m}$ . The XPS results showed that the films were mainly titanium dioxide. Some coke-like deposits, which may contain or be changed by OH, NH, C–C, C–O, and C=O groups, were doped in the films. The XRD results

showed that the films exhibited an amorphous structure with a small amount of anatase and rutile phase. The crystal phase was detected by Raman spectroscopy. Researches about the corrosion resistance of the films are in progress in our further study.

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