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Materials

Effect of SiO₂ addition on the dielectric properties and microstructure of BaTiO₃-based ceramics in reducing sintering

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Abstract: The effect of SiO₂ doping on the sintering behavior, microstructure, and dielectric properties of BaTiO₃-based ceramics has been investigated. Silica was added to the BaTiO₃-based powder prepared by the solid state method with 0.075mol%, 0.15mol%, and 0.3mol%, respectively. The SiO₂-doped BaTiO₃-based ceramic with high density and uniform grain size were obtained, which were sintered in reducing atmosphere. A scanning electron microscope, X-ray diffraction, and LCR meter were used to determine the microstructure as well as the dielectric properties. SiO₂ can form a liquid phase belonging to the ternary system of BaO-TiO₂-SiO₂, leading to the formation of BaTiO₃ ceramics with high density at a lower sintering temperature. The SiO₂-doped BaTiO₃-based ceramics can be sintered to a theoretical density higher than 95% at 1220°C with a soaking time of 2 h. The dielectric constants of the sample with 0.15mol% SiO₂ addition sintered at 1220°C is about 9000. Doping with a small amount of silica can improve the sintering and dielectric properties of BaTiO₃-based ceramics.

Key words: barium titanate; multilayer ceramic capacitor (MLCC); reducing atmosphere; dielectric properties

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1. Introduction

Barium titanate (BaTiO₃, BT), a perovskite structure, has been widely investigated because of its dielectric and ferroelectric properties [1-2]. It is obvious that replacing the noble metal with base metal in multilayer ceramic capacitors (MLCC) can significantly reduce the production costs. The best candidate for such a replacement is Ni. Dielectric formulation and process technology for MLCC has been extensively investigated in the electronics industry [3]. Recently, MLCC with base-metal internal electrodes such as nickel have been developed to reduce the material cost [4-5]. In this case, dielectric should be sintered in a reducing atmosphere to prevent the oxidation of internal electrode. However, barium titanate ceramic becomes semiconducting and loses their high insulation resistance due to the unlocalized electrons produced by the formation of oxygen vacancies in a reducing atmosphere [6]. For the fabrication of MLCC with base metal electrodes, Mn is often added to maintain high insulation resistance even after sintering in a reducing atmosphere [7].

The Ni electrode is easily oxidized during the firing process. Consequently, low oxygen partial pressure during firing is indispensable. A reducing atmosphere may cause the formulation of oxygen vacancies in the ceramic body of a capacitor. For that reason, annealing as the sequential process of sintering was carried out in order to oxidize the ceramic bodies. The oxygen partial pressure was controlled by the equilibrium of H_2/H_2O , *i.e.* the water vapor pressure was adjusted by bubbling a hydrogen/nitrogen mixture through water at a controlled temperature. On the other hand, the reduction of the sintering temperature of BaTiO₃-based ceramics is very necessary to enable it to co-fire with Ni or Cu electrodes and for better electrical performance. The addition of some glass components is sometimes useful in decreasing the ceramic sintering temperature [8]. In the conventional method, glass was prepared in advance and then mixed with the other starting materials [9-10].

The purpose of this study is to investigate the ef-

2. Experiment

tion.

The samples were prepared from high purity BaCO₃, TiO₂, CaCO₃, ZrO₂, and MnO₂ raw materials by using the conventional solid state reaction method. The raw materials are shown in Table 1. The average particle sizes for all raw materials are listed in Table 2. The SiO₂-doped powder was pressed into disk form (10 mm in diameter, 1 mm in thickness) using the 3wt% PVA binder. The specimens were sintered at the temperatures ranging from 1220 to 1300°C for 2 h in reducing atmosphere (o₂: 50 ppm). The densities of the sintered compacts were measured by the Archimedes' method using water as the liquid medium. The microstructures of the polished samples were examined by SEM (model S2500, Hitachi, Tokyo, Japan). Using an automatic capacitance bridge (HP4278A), the capacitance and dissipation factor of the samples were measured at 1 kHz/1 V, and the temperature dependence of relative capacitance was studied in the range of -30 to 85°C. Insulation resistance was measured using HP 4140A after applying 50 V D.C.

using scanning electron microscopy (SEM) examina-

Table 1. Brand of the chemical agents

Reagent	Molecular weight	Brand
BaCO ₃	197.35	Nippon-chemical
CaCO ₃	100.08	Merck
TiO ₂	79.9	Fuji-titanium
ZrO_2	123.22	Merck
MnO ₂	86.94	Merck

 Table 2.
 Average particle size distribution of raw materials

Raw powder	Average particle size, D_{50} / µm
BaCO ₃	1.75
CaCO ₃	0.46
TiO ₂	0.29
ZrO_2	0.27
MnO_2	1.50

3. Results and discussion

3.1. Effect of sintering temperature and SiO₂ on the physical characteristic and microstructure

The particle size of the starting powder was controlled at $0.4\pm0.1 \ \mu\text{m}$. The curves of the bulk density of BaTiO₃-based ceramic pellets *vs.* sintering temperature with different SiO₂ additions are shown in Fig. 1. It can be seen that the density of the sintered ceramics without SiO₂ addition increases approximately linearly with sintering temperature ranging from 1220 to 1300°C. The bulk density of Ba-TiO₃-based ceramics sintered at 1300°C for 2 h is 5.5 g/cm³. The bulk density can be increased significantly at 1220°C by 0.15mol% and 0.3mol% SiO₂ addition. It is well known that the addition of some glass components is sometimes useful in decreasing the ceramic sintering temperature [8].

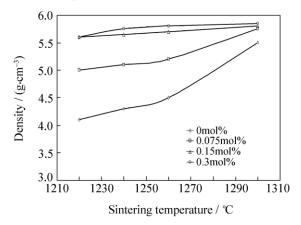


Fig. 1. Bulk density vs. sintering temperature of the Ba-TiO₃-based ceramics with different SiO₂ additions.

The X-ray diffraction spectra of the as-sintered ceramic are given in Fig. 2. The XRD pattern of the sintered sample mainly consists of the BaTiO₃ crystalline phase without minority phases in the ceramic. Fig. 3 shows the SEM photographs of the BaTiO₃ samples sintered at 1220°C for various SiO₂ additions. Many pores in the sample can be observed in Fig. 3(a), which was sintered at 1220°C without SiO₂ addition. Significant densification and grain growth progress as the SiO₂ content increases. Fig. 4 shows the grain size of the BaTiO₃-based specimen as a function of the SiO₂ content. The sintered specimen with 0.15mol% SiO₂ addition exhibits the average grain size of about 2.5 μ m, where the bulk density is 5.7 g/cm³ (> 95% of the theoretical density). However, the sintered specimen with 0.075mol% SiO₂ addition shows smaller grain sizes with poor sintering densities (83% of the theoretical density). This result represents the normal densification process of ceramic bodies, indicating that there is no densification without grain growth. It was reported that the driving forces for densification and grain growth are comparable in magnitude, both being proportional to the reciprocal grain size in this final-stage sintering [11-12]. Thus, this has greatly hampered the efforts to produce dense materials with a nano-size grain structure. For the SiO₂ addition from 0.15mol% to 0.3mol% and sintered at 1220°C, a slight grain growth was observed, whereas the density insignificantly increased still keeps at 5.7 g/cm³.

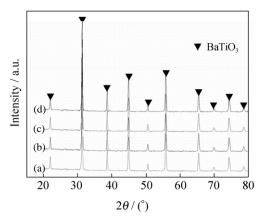


Fig. 2. X-ray diffraction spectra of the BaTiO₃-based ceramics at 1220°C with different SiO₂ additions: (a) 0; (b) 0.075mol%; (c) 0.15mol%; (d) 0.3mol%.

3.2. Effects of sintering temperature and SiO_2 on dielectric properties

The dielectric constant of the BaTiO₃-based ceramics was measured at 1 kHz, and the result is shown in Fig. 5. Similar to the bulk density *vs*. sintering temperature curve of the sample without SiO₂ addition in Fig. 1, the dielectric constant of the BaTiO₃-based ceramics increases with increasing sintering temperature. The ε_r value of the BaTiO₃-based ceramic without SiO₂ addition spans from 5600 (1220°C) to 15000 (1300°C). For this class of ceramics, it is possible to obtain high dielectric constant, e.g. $\varepsilon_r > 12000$, only by raising the sintering temperature above 1300°C. The sintering temperature can be lowered down to 1220°C by an addition of 0.15mol% or 0.3mol% of SiO₂ glass to the ceramics. Therefore, it can be concluded that the SiO₂ glass phase is effective in lowering the sintering temperature of the BaTiO₃-based ceramics having high dielectric constant. The loss tangent of the BaTiO₃-based ceramics is shown in Fig. 6. The dielectric loss of BaTiO₃-based ceramics is apparently reduced by enhancing the sintering temperature, and it also depends on the addition of SiO₂. For the Ba-TiO₃-based ceramics without SiO₂ addition, it is closely related to the sintering temperature only, e.g., the dielectric losses of the ceramic are 5.44%, 3.62%, 2.43%, and 2.33% for sintering at 1220, 1240, 1260, and 1300°C, respectively.

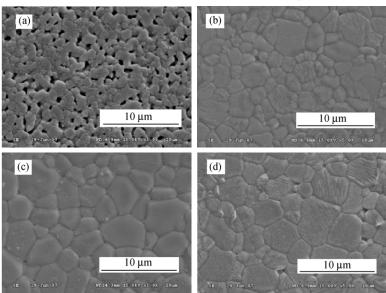


Fig. 3. SEM images of the samples of the BaTiO₃-based ceramics sintered at 1220°C with different SiO₂ additions: (a) 0; (b) 0.075mol%; (c) 0.15mol%; (d) 0.3mol%.

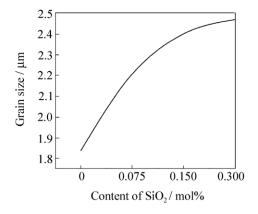


Fig. 4. Relationship between the mean grain size and SiO_2 addition for the BaTiO₃-based ceramics sintered at 1220°C.

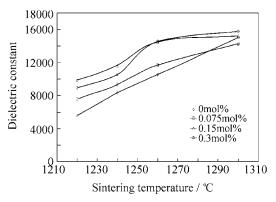


Fig. 5. Curves of the dielectric constant vs. sintering temperature of BaTiO₃-based ceramics with different SiO₂ contents.

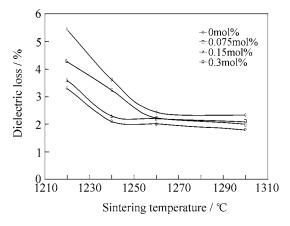


Fig. 6. Curves of the dielectric loss *vs.* sintering temperature of BaTiO₃-based ceramics with different SiO₂ contents.

Fig. 7 shows the insulation resistance of Ba-TiO₃-based ceramics sintered at 1220°C as a function of SiO₂ content. It can be seen that the insulation resistance of BaTiO₃-based ceramics increases with increasing SiO₂ content. The reason for this can be explained by the density of ceramics. Fig. 8 shows the temperature coefficient of the capacitance (TCC) of BaTiO₃-based ceramics at the temperature ranging from -30 to 85°C. All the samples can satisfy the Y5V TCC specification ($\Delta C/C$: +20% ~ -80%).

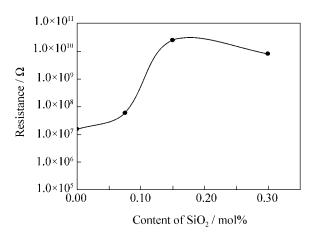


Fig. 7. Insulation resistance of BaTiO₃-based ceramics sintered at 1220°C as a function of SiO₂ content.

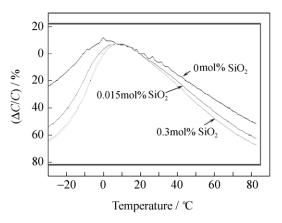


Fig. 8. TCC curves of the BaTiO₃-based ceramics with different SiO_2 additions.

4. Conclusions

(1) The grain size is enhanced by increasing the SiO_2 content for the BaTiO₃-based ceramics sintered at 1220°C in reducing atmosphere.

(2) Sintering temperatures can be lowered from 1300 to 1220°C when SiO_2 is added at 0.15mol% and 0.3mol%, respectively.

(3) The SiO₂ dopant plays the role of sintering in aiding to enhance the densification of BaTiO₃-based ceramics without change in the stoichiometry. Better electrical properties are obtained with increasing SiO₂ content due to the improvement of densification.

References

- H. Kishi, Y. Mizuno, and H. Chazono, Base-metal electrode-multilayer ceramic capacitors: past, present and future perspectives, *Jpn. J. Appl. Phys.*, 42(2003), No.1, p.1.
- [2] T. Li, L.T. Li, Y. Kou, and Z.L. Gui, Stable temperature dependence of dielectric properties in Ba-TiO₃-Nb₂O₅-Co₃O₄-Gd₂O₃ system, *J. Mater. Sci. Lett.*, 19(2000), No.11, p.995.
- [3] J.M. Herbert, Method of manufacture, [in] *Ceramic Dielectric and Capacitor*, Northamptonshire, UK, 1985, p.84.
- [4] H. Shizuno, S. Kusumi, H. Saito, and H. Kishi, Properties of Y5V multilayer ceramic capacitors with nickel electrodes, *Jpn. J. Appl. Phys.*, 32(1993), p.4380.
- [5] R.Z. Chen, X.H. Wang, H. Wen, L.T. Li, and Z.L. Gui, Enhancement of dielectric properties by additions of Ni nano-particles to an X7R-type barium titanate ceramic matrix, *Ceram. Int.*, 30(2004), p.1271.
- [6] Y.H. Han, J.B. Appleby, and D.M. Smyth, Calcium as an acceptor impurity in BaTiO₃, *J. Am. Ceram. Soc.*, 70(1987), No.2, p.96.
- [7] I. Burn, Mn-doped polycrystalline BaTiO₃, J. Mater. Sci., 14(1979), p.2453.
- [8] N. Wang, M.Y. Zhao, and Z.W. Yin, Low-temperature firing in microwave dielectric ceramic, *J. Inorg. Mater.*, 17(2002), No.5, p.915.
- [9] D.-W. Kim, D.-G. Lee, and K.S. Hong, Low-temperature firing and microwave dielectric properties of BaTi₄O₉ with Zn-B-O glass system, *Mater. Res. Bull.*, 36(2001), p.585.
- [10] H. Jantunex and R. Rautioaho, Compositions of MgTiO₃-CaTiO₃ ceramic with two borosilicate glasses for LTCC technology, J. Eur. Ceram. Soc., 20(2000), p.2331.
- [11] C.P. Cameron and R. Raj, Grain-growth transition during sintering of colloidally prepared alumina powder compacts, *J. Am. Ceram. Soc.*, 71(1988), No.12, p.1031.
- [12] V.V. Skorokhod and A.V. Ragulya, Features of nanocrystalline structure formation on sintering of ultra-fine powders, [in] *Nanostructure Materials*, Kluwer Academic Publishers, The Netherlands, 1998, p.387.